

Copyright © 2012 American Scientific Publishers All rights reserved Printed in the United States of America Journal of Nanoscience and Nanotechnology Vol. 12, 1–3, 2012

Superconductivity of Nanostructured Pb₇Bi₃ Films Doped by Ce

V. S. Stolyarov^{1, *}, V. N. Zverev¹, E. Y. Postnova¹, G. V. Strukov¹, G. K. Strukova¹, A. Y. Rusanov², and I. M. Shmitko¹

¹ Insitute of Solid State Physics RAS, Academika Osipyana, 2, Chernogolovka, Moscow reg., Russia ² JSC "Applied Radiophysics," Severniy pr. 1, Chernogolovka, Moscow reg., Russia

By means of electrochemical deposition from electrolytes, containing salts of Pb and Bi (0.03 mol/l and 0.02 mol/l respectively) thin films of intermetallic Pb_7Bi_3 have been fabricated. The superconducting transition temperature of the films was measured to be around 7.8 K. The deposition of the films with thickness of 50–100 nm was performed via passing rectangular current pulses with given amplitude and length. It was shown that adding salt of Ce into the electrolyte leads to a significant growth of the T_c for the deposited films reaching its maximum at the salt concentration of 0.06 mol/l. X-ray analysis data revealed the single phase of Pb_7Bi_3 films with hexagonal structure (SG) having a textures parallel to (101) plane. The morphology of the film surface is characterized by nanocluster structure with typical grain size around 70–80 nm. For the films, fabricated with adding salt of Ce, together with the intermetallic phase of Pb_7Bi_3 , the second phase containing Bi is detected. At the same time, the typical grain size is reduced to 20–30 nm. Additionally, the suppression of the superconductivity in the grown films is investigated. The influence of the composition and structure on the superconducting critical temperature is discussed for both types of the fabricated films.

Keywords: Superconductivity, Nanostructured, Pb₇Bi₃ Films, Electrodeposition.

At present times, vacuum sputtered layers of metallic superconductors (Nb, Al, Pb, MoGe etc) are used as the superconducting elements of new electronic schemes.¹ In some of the tasks related to the use of normal metals a method of electrodeposition from solution is implemented. A good example of that is growing nanosized threads in porous matrix,² or creating multilayer structures with many interchanging layers.³ Many technological problems connected with the technique of superconducting materials vacuum sputtering could have been avoided if the method of electrodeposition from solution was used instead. However, electrodeposition of Nb layers from solutions under regular conditions is still an unaccomplished task, and frequently used lead layers with nanometer thickness oxidize easily, losing their superconducting properties. At the same time intermetallic Pb7Bi3 is more resistant to oxidation and has a superconducting transition temperature $T_c = 7.8$ K, yet the known methods of electrodeposition of PbBi alloy using water electrolytes do not guarantee any reproducibility of layers phase composition and properties. Our approach to obtaining metallic alloy layers consists of using electrolytes on the basis of complex-forming aproton-dipolar solvent, which allows us to obtain films from alloys (including intermetallic compounds ones) both in electroless process⁴ and using the method of pulse electroplating.⁵

The distinctive feature of thin films obtained under such conditions is their nanocrystalline structure. The main goal of the present study is the development of the PbBi electrodeposition techniques of nanocrystalline films and investigation of their superconducting properties.

PbBi films with the characteristic thickness of 20–100 nm were grown from the solution on brass and copper substrates using pulse electroplating. The sample of 5 cm² was used as a cathode whereas platinum foil had the role of the anode. Rectangular current pulses were sent through the electrolyte solution, the amplitude of 100–400 mA and pulse length 3–50 msec were controlled via density was varied in the range of 50–100 mA/cm2. The electrolyte contained ions of Pb²⁺, Bi³⁺ or Pb²⁺, Bi³⁺ and Ce³⁺ in organic aproton- dipolar solvent.⁵

^{*}Author to whom correspondence should be addressed.

J. Nanosci. Nanotechnol. 2012, Vol. 12, No. xx



Fig. 1. The diffraction pattern of Pb/Bi-71/29 at.% film, grown on brass substrate by pulse electroplating. The percentages in the peaks indicate the intensity of the reflections.⁶

The organic aproton-dipolar solvent in combination with ammonium chloride provides the complex-forming process in the electrolyte, which allows obtaining thin films of PbBi and PbBiCe alloys. The amount of Pb and Ce in the grown films was adjusted by changing the current density as well as the concentration of metal ion in the solution. The intermetallic Pb7Bi3 was grown from the electrolyte containing 0.03 mol/l of Pb²⁺ and 0.02 mol/l of Bi³⁺. The current density and electrolyte temperature during the electrodeposition were 70–100 mA/cm² and 60 °C respectively. During the deposition the electrolyte was also carefully agitated.

The transport and superconducting properties of the obtained films were measured using standard 4-point scheme. The resistance of the films was examined at different temperatures from 300 K down to the superconducting

transition around 7.8 K and then the critical field Hc_2 was measured at 4.2 K. SEM Supra V50 and Siemens D500 systems were used for the investigation of the film composition, morphology and X-ray analysis (Cu K α -radiation) respectively.

Stolyarov et al.

Figure 1 shows the diffraction pattern of Pb/Bi-71/29 at.% film, grown on brass substrate by e by pulse electroplating, the film thickness is 80 nm. The X-ray analysis reveals the single phase surface Pb_7Bi_3 , which has a hexagonal close-packed structure (SG.P63 mmc) having the following lattice parameters a = 3.5058 Å, c = 5.7959 Å respectively, and texture parallel to (101) plane. The film surface morphology is presented in Figure 2.

The dependence of the film resistance versus temperature is quite predictable, see Figure 3(a), and demonstrates the superconducting transition around $T_c = 7.8$ K, which agrees well with the data for bulk intermetallic Pb₇Bi₃. The PbBi film doping with Ce was performed by adding sols of Ce into the electrolyte. The maximal critical temperature Tc of 10.3 K (Fig. 3(b)) was obtained at Ce salt concentration around 0.06 mol/l, while the additional increase of the salt concentration does not result in further T_c growth.

At 4.2 K, well below the superconducting transition, the critical field Hc_2 is reached around 7kOe if the external magnetic field is applied perpendicular to the film surface (Fig. 4).

The composition of the Pb/Bi-71/29 at.% film with $T_c = 10.3$ K was determined. The diffraction pattern indicate the presence of Pb₇Bi₃ phase and around 10% of a secondary phase with the central reflex corresponding to Bi (T_c for Bi 6.17 K therefore the basic contribution to superconducting transport properties is done by phase Pb7Bi3).



Fig. 2. The film surface (a) without Ce; (b) with Ce. The grain size around 20 nm.



Fig. 3. (a) Temperature dependence of resistance for Pb_7Bi_3 film. (b) Temperature dependence of resistance for Pb-Bi+Ce film.

The morphology of the film surface is characterized by the nanocluster 20–30 nm grain structure (Figs. 2(a, b)), which indicates a significant grain size decrease, see Figure 2 for comparison. Such structure formation is obviously caused



Fig. 4. Field dependance of resistance for Pb-Bi+Ce film.

by adding Ce atoms. A T_c increase for the film doped with Ce can be related to its nanocluster structure. It was previously theoretically predicted that the critical temperature increase can occur in metallic ordered nanocluster systems.⁷ The influence of Ce doping on superconducting properties of electrodeposited PbBi films is not entirely clear, in particular the film nanoclustered structure should be studied in more detail.

References and Notes

- 1. A. A. Bannikh, J. Pfeiffer, and V. S. Stolyarov, *Phys. Rev. B* 79, 054501 (2009).
- 2. X.-Y. Sun, F.-Q. Xu, and Z.-M. Li, *Mater. Chem. Phys.* 90, 1 (2005).
- G. V. Strukov, G. K. Strukova, E. D. Shoo, S. I. Bozhko, and Y. P. Kabanov, Pribory I Teknika Experimenta 123 (2009); Instruments and Experimental Techniques 52, 727 (2009).
- 4. G. K. Strukova, G. V. Strukov, and I. E. Batov, *Mater. Chem. Phys.* 119, 3 (2010).
- 5. G. K. Strukova, G. V. Strukov, and S. I. Bozhko, *Journal of Nanoscience and Nanotechnology* 11, 1 (2011).
- 6. S. Rasmussen and B. Lundtoft, Powder Diffraction 2, 28 (1987).
- 7. V. Z. Kresin and Y. N. Ovchinnikov, Phys. Usp. 51, 427 (2008).

Received: 1 December 2010. Accepted: 1 May 2011.